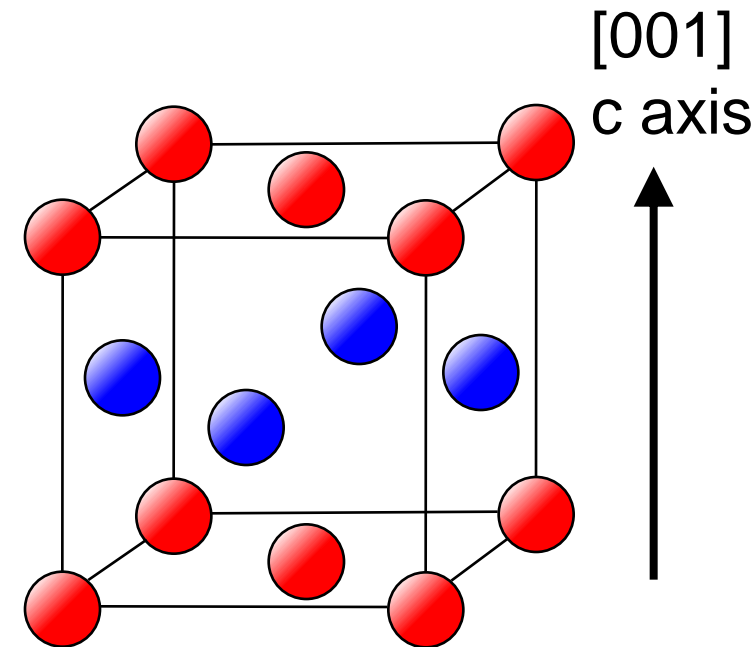


# Self- and Impurity Diffusion in $\gamma$ -TiAl Single Crystals

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*$\gamma$ -TiAl intermetallic compounds*



L1<sub>0</sub>-type structure

**Investigate mechanism  
of atomic diffusion**

**Diffusion anisotropy**

a-axis direction :  $D_a$   
c-axis direction :  $D_c$



**using single crystals**

**Self- diffusion : Ti, In(Al)  
Impurity diffusion: Fe, Ni**

# Experimental procedure

## Specimen preparation

Arc-melting

Single crystal growth

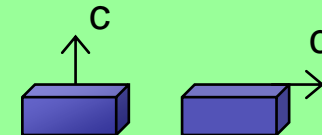
Floating Zone method

Anneal for homogeneity

1353 K, 72 h

Cutting & polishing

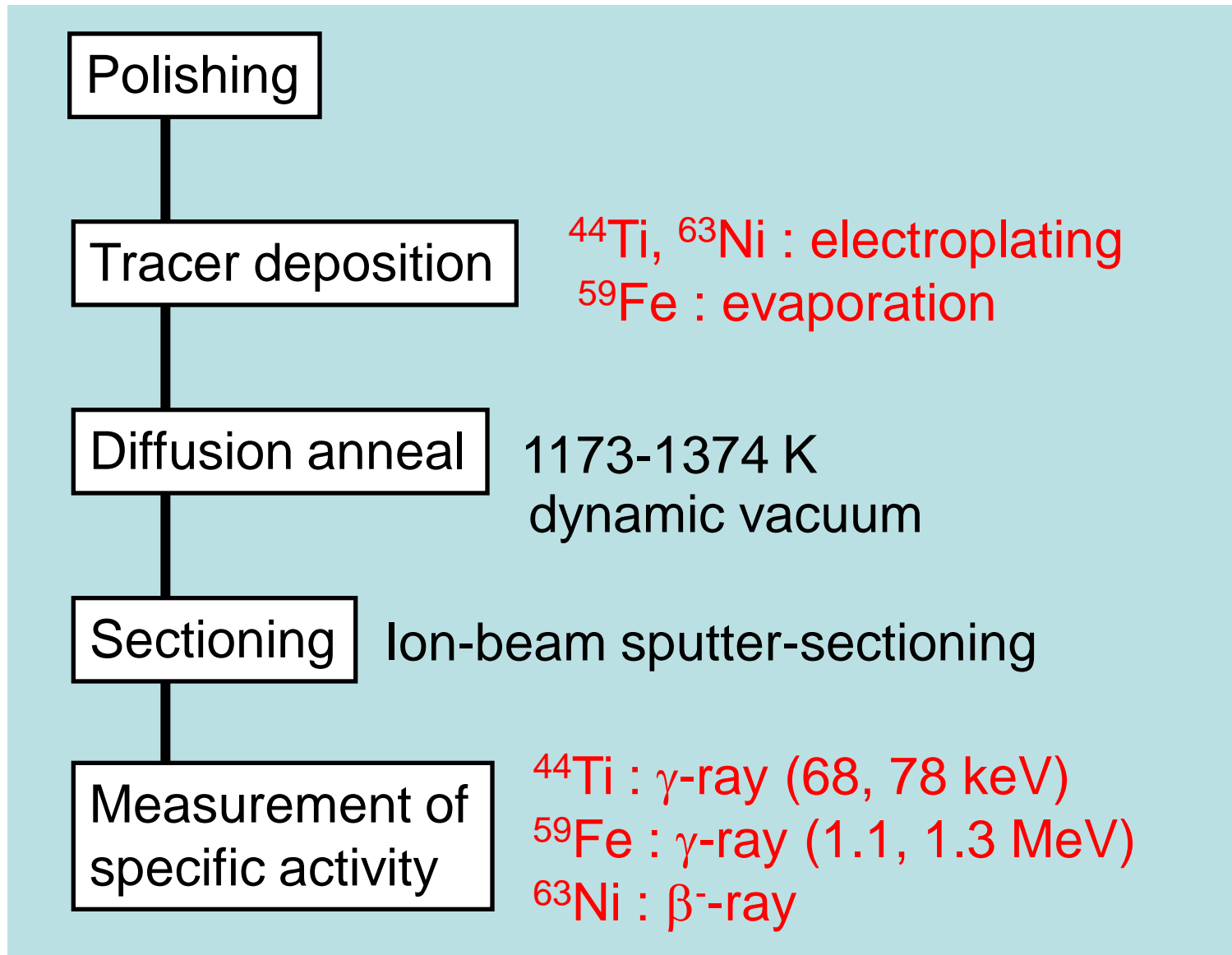
two directions



Composition analysis

53.2~53.6 mol%Al

Tracer experiments using radioisotopes *Ti, Fe, Ni in TiAl*



## Tracer experiments using SIMS *In in TiAl*

Polishing

Ion implantation

$^{115}\text{In}$  150 keV  
 $1 \times 10^{17-18}$  ions/cm<sup>2</sup>

Diffusion anneal

1211-1373 K

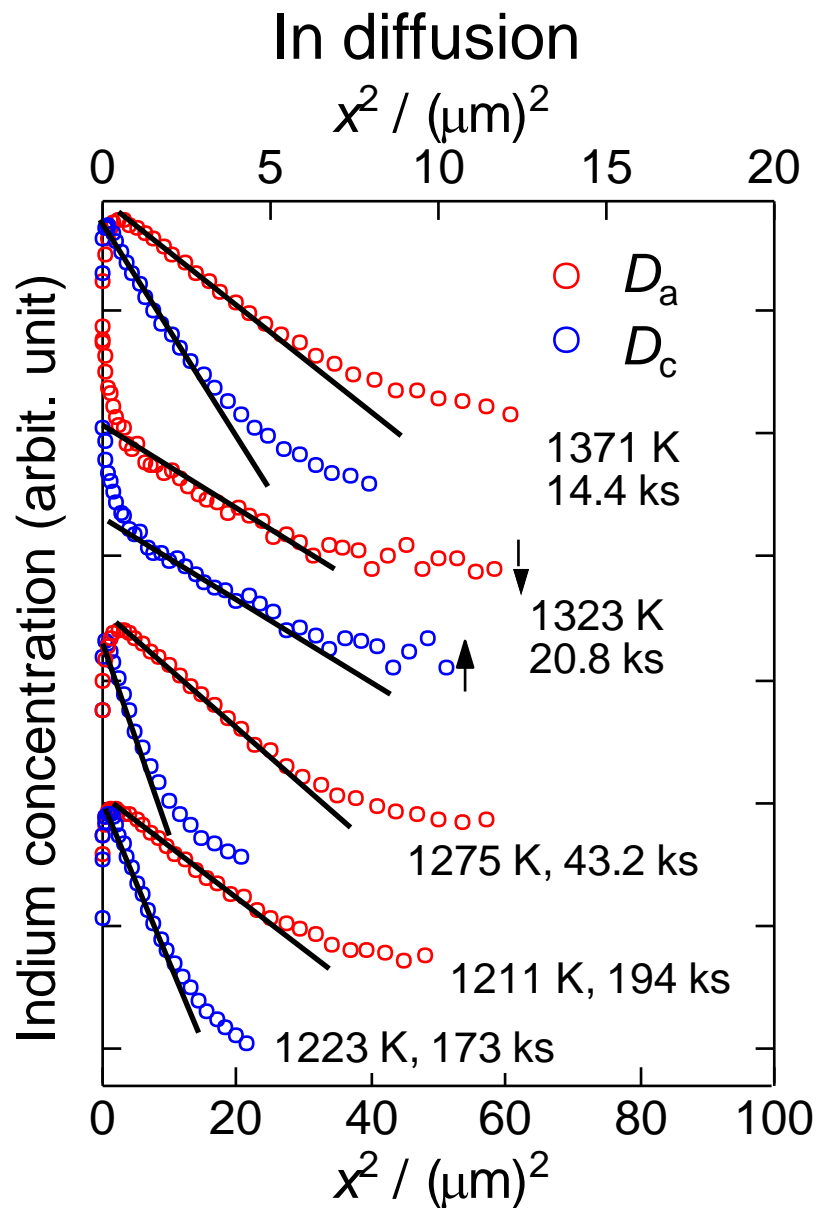
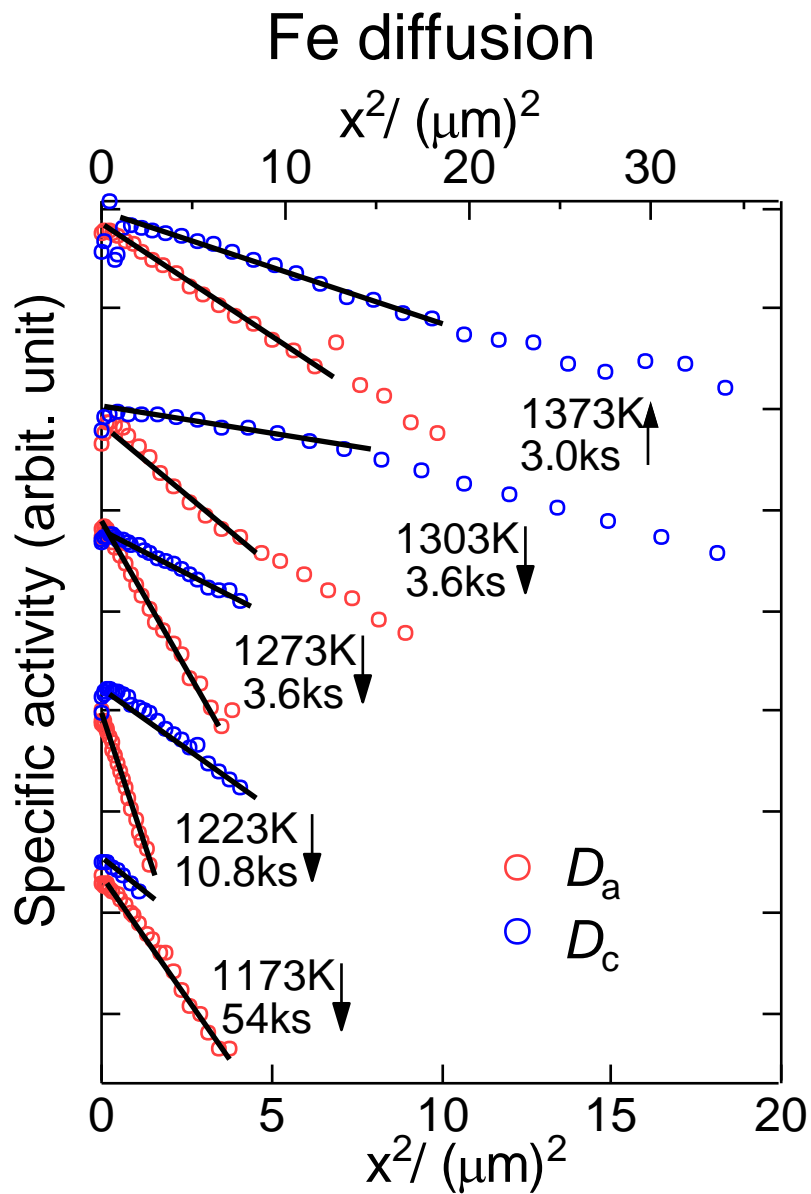
In a quartz capsule

- Ar gas atmosphere
- Indium balls (0.04g)

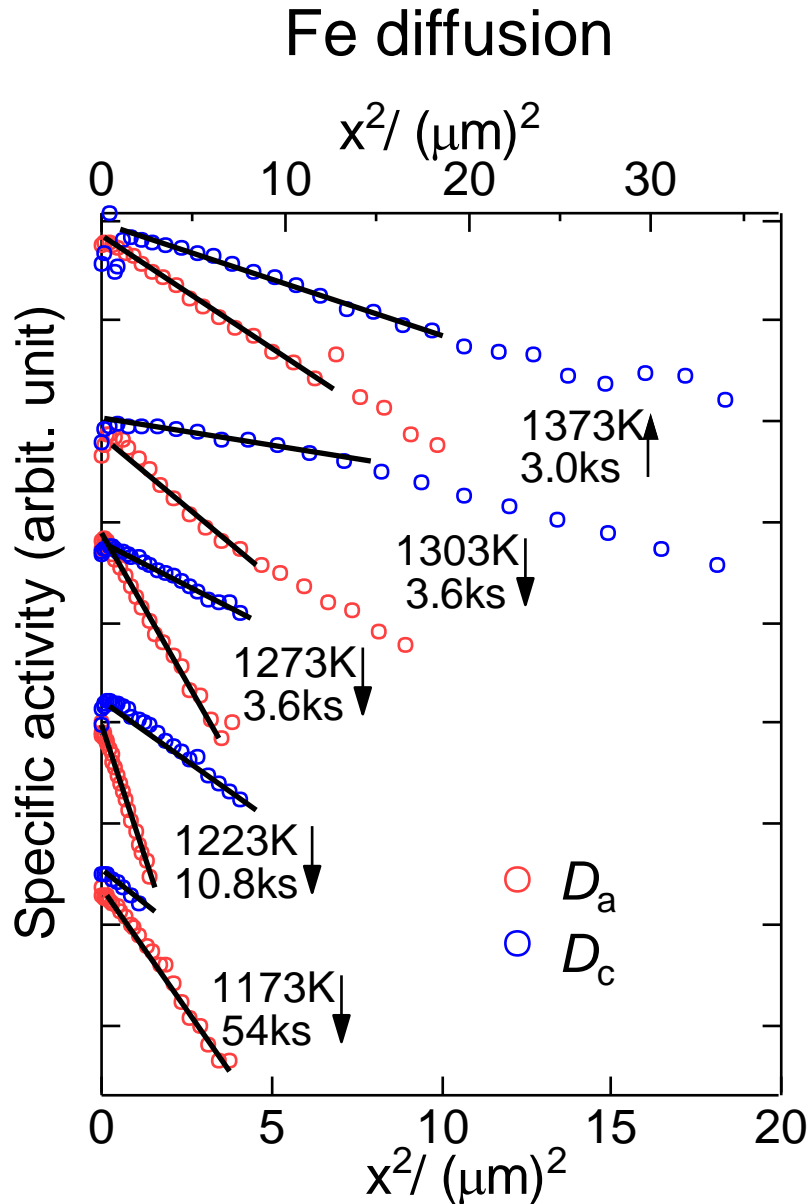
SIMS analysis

$\text{O}_2^+$ , 10keV

# Typical penetration profiles of diffusion in TiAl



# Calculation of diffusion coefficient



## Thin film solution

$$c = \frac{\alpha}{2\sqrt{\pi Dt}} \exp\left(-\frac{x^2}{4Dt}\right)$$

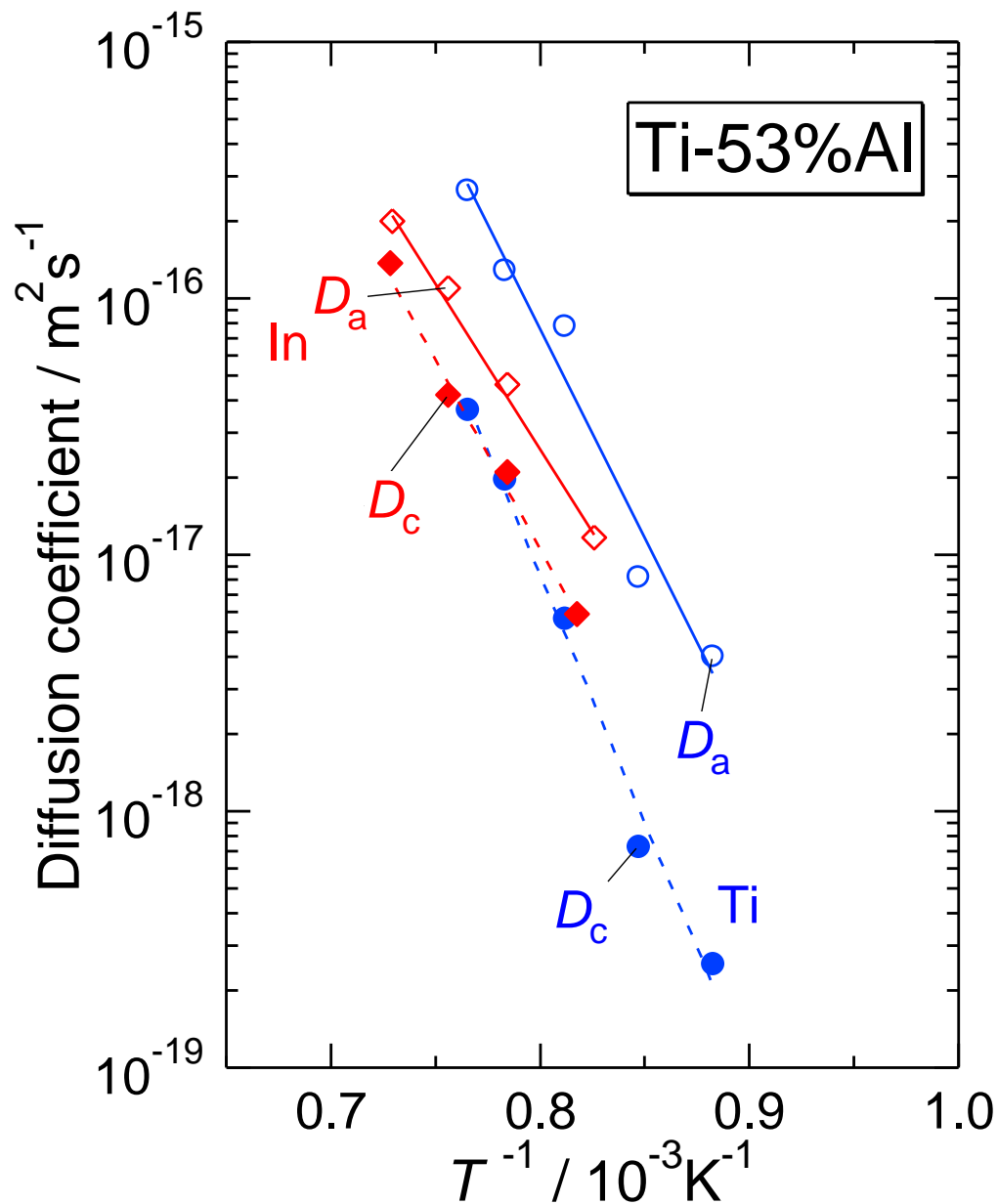
$D$  : diffusion coefficient

$x$  : distance

$t$  : annealing time

$$\ln c = \ln a - \frac{x^2}{4Dt}$$

# Diffusion of Ti and In in TiAl single crystal



## Anisotropy

	$D_c / D_a$
Ti	0.07-0.15
In	0.38-0.69

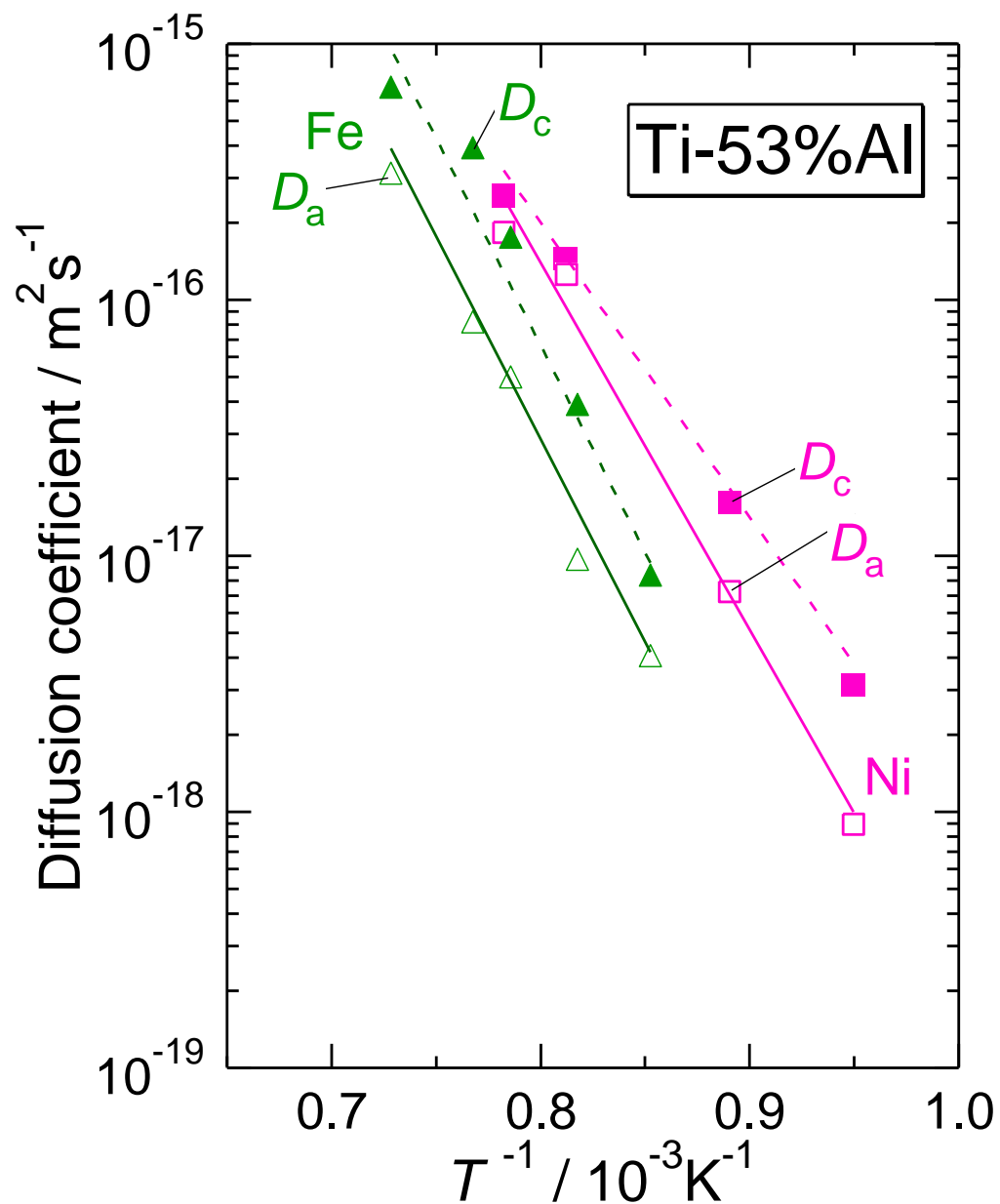
## Activation energy

	$Q_c$	$Q_a$
Ti	$3.85 \pm 0.23$	$3.22 \pm 0.36$
In	$2.95 \pm 0.20$	$2.57 \pm 0.14$

eV



# Diffusion of Fe and Ni in TiAl single crystal



## Anisotropy

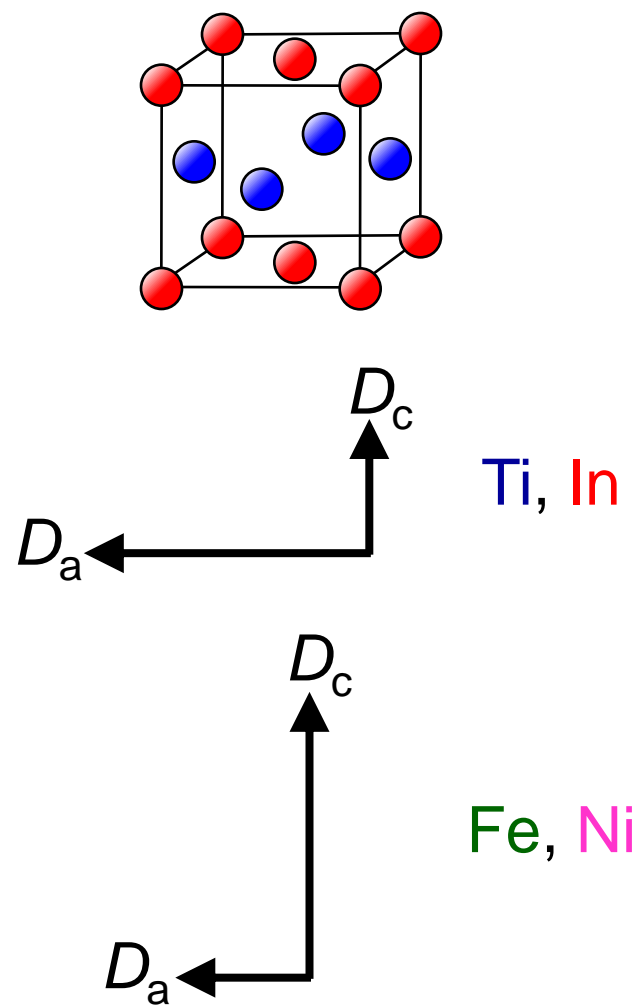
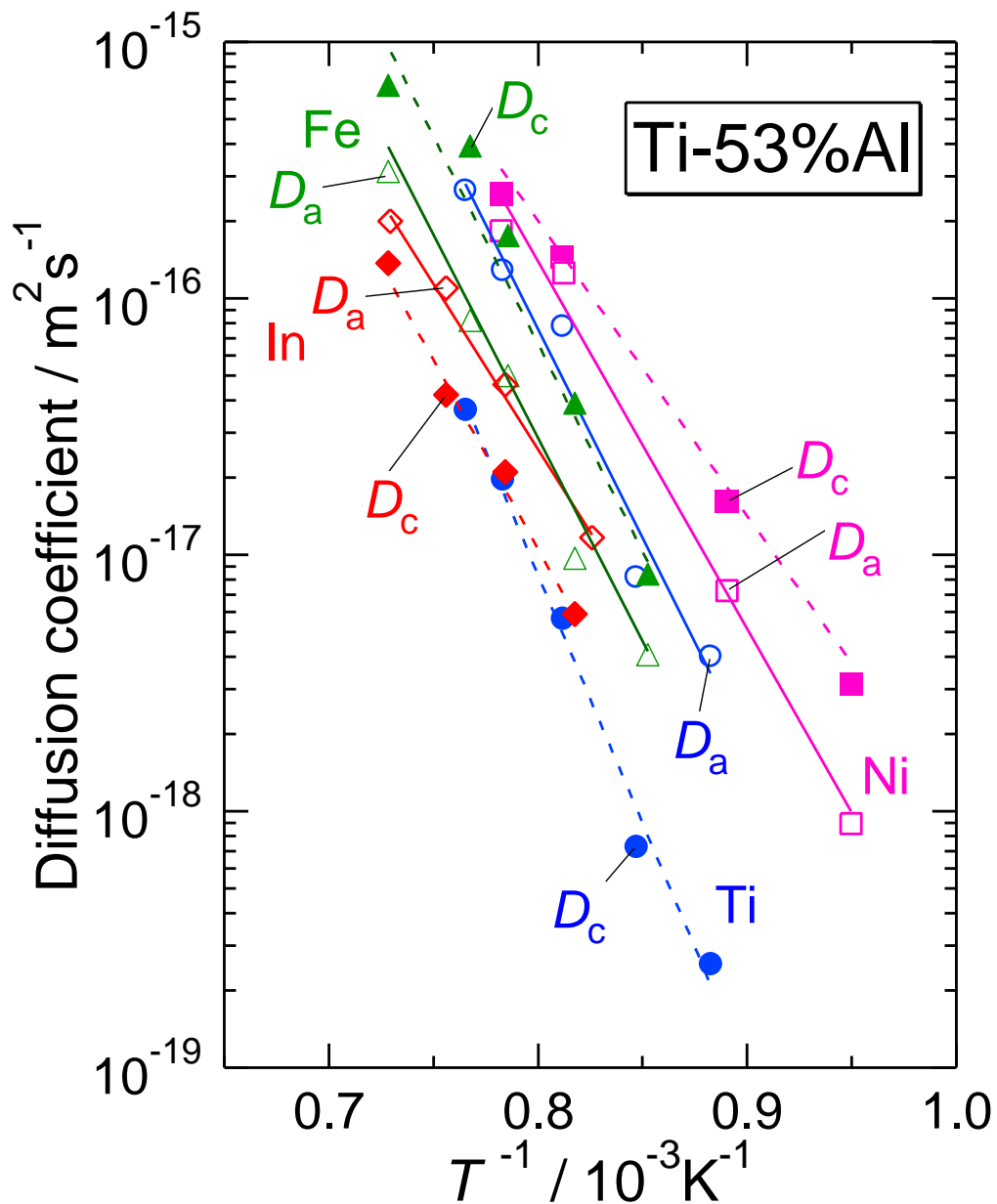
	$D_c / D_a$
Fe	1.55-4.09
Ni	1.40-3.51

## Activation energy

	$Q_c$	$Q_a$
Fe	$3.21 \pm 0.45$	$3.14 \pm 0.31$
Ni	$2.31 \pm 0.07$	$2.84 \pm 0.21$

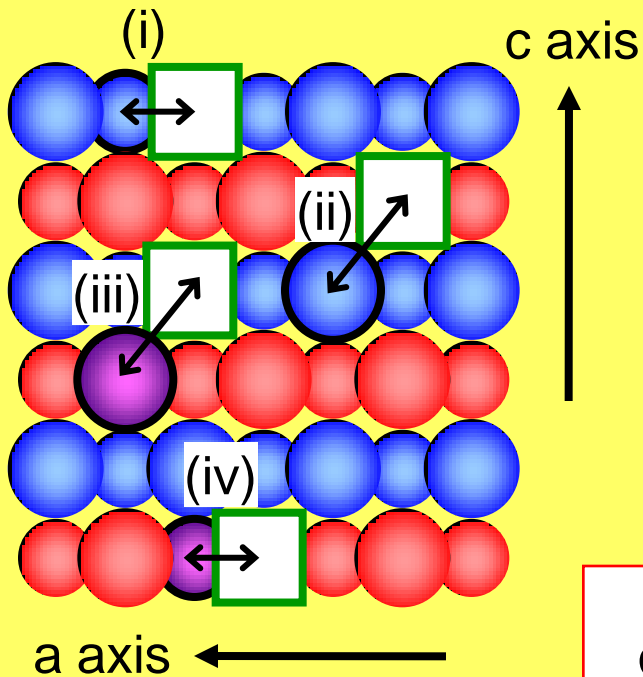
eV

# Anisotropy of diffusion in TiAl single crystal



**Why ???**

# Expression of diffusion coefficient for $L1_0$ -ordered alloys



## Type of jumps of A atoms

- (i) Jumps on the planes for A atoms
- (ii) Jumps from the A atom planes to B atom planes
- (iii) Reverse jumps of (ii)
- (iv) Jumps on the plane for B atoms as antisite atoms.

### c-axis direction

$$D_c = D_c(\text{ii}, \text{iii})$$

### a-axis direction

$$D_a = D_a(\text{i}) + D_a(\text{ii}, \text{iii}) + D_a(\text{iv})$$

# Activation energy for Ti diffusion in TiAl single crystal

c-axis direction

$$D_c = D_c(\text{ii+iii})$$

3.01-3.04

※ Calculated for  $\text{Ti}_{46}\text{Al}_{54}$  by Prof. Y. Mishin using embedded atom method. (The unit is eV.)

a-axis direction

$$D_a = D_a(\text{i}) + D_a(\text{ii+iii}) + D_a(\text{iv})$$

2.40    3.01-3.04    3.06

Type (i) is predominant

Ti atoms jump on the Ti sublattice!

Calculation

$$Q_{\perp} < Q_{\parallel}$$

2.4    3.0



Experiments

$$Q_{\perp} < Q_{\parallel}$$

3.2    3.9

## Activation energy for Al diffusion in TiAl single crystal

a-axis direction

$$D_c = D_c(\text{ii, iii})$$

2.60-2.70

c-axis direction

$$D_a = D_a(\text{i}) + D_a(\text{ii, iii}) + D_a(\text{iv})$$

2.84

2.60-2.70

2.14

Type (iv) is predominant

**Al atoms jump on the Ti sublattice!**

Al diffusion  
(calculation)

$$Q_a < Q_c$$

2.1 2.7



In diffusion  
(experiments)

$$Q_a < Q_c$$

2.6 3.0

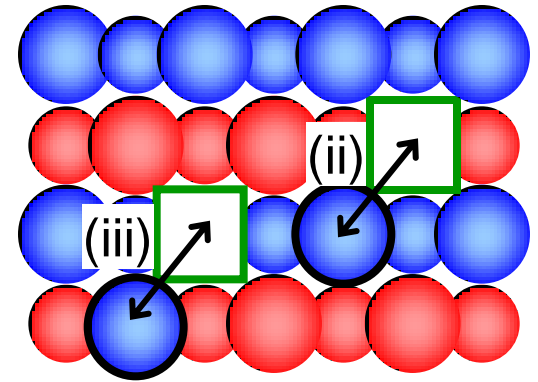
## Diffusion mechanism of Fe in TiAl

c-axis direction

$$D_c = D_c(\text{ii}, \text{iii})$$

a-axis direction

$$D_a = D_a(\text{i}) + D_a(\text{ii}, \text{iii}) + D_a(\text{iv})$$



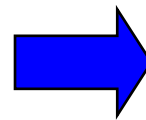
Assuming that  $D_a(\text{ii}, \text{iii})$  is predominant,

$$D_c(\text{ii}, \text{iii}) = 4 D_a(\text{ii}, \text{iii}) \quad \rightarrow \quad D_a \leq 4D_c \quad \text{and} \quad Q_a \approx Q_c$$

Fe diffusion

$$D_c / D_a = 2.2-4.8$$

$$Q_c \approx Q_a$$



Fe atoms diffuse  
on both Ti and Al  
sublattices !

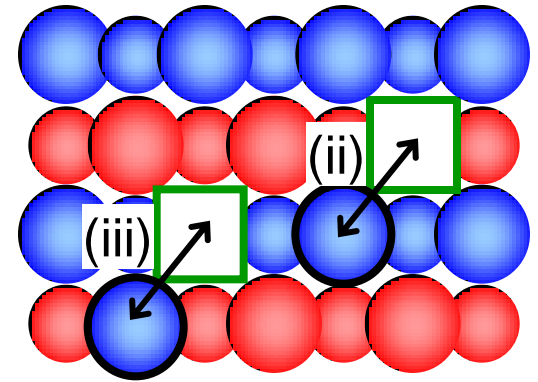
# Diffusion mechanism of Ni in TiAl

c-axis direction

$$D_c = D_c(\text{ii, iii})$$

a-axis direction

$$D_a = D_a(\text{i}) + D_a(\text{ii, iii}) + D_a(\text{iv})$$



Assuming that  $D_a(\text{ii, iii})$  is predominant,

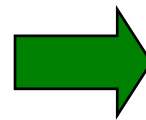


$$D_a < 4D_c \text{ and } Q_a \approx Q_c$$

Ni diffusion

$$D_c / D_a = 1.4-3.6$$

$$Q_c < Q_a$$



Type (ii, iii) may be not predominant !



Mixed mechanism??

## Conclusions

### Ti and In in TiAl

$$\text{Ti: } D_c / D_a = 0.07-0.15 \quad Q_c > Q_a$$

$$\text{In: } D_c / D_a = 0.38-0.69 \quad Q_c > Q_a$$

Ti atoms jump on Ti sublattice.

In atoms jump on Ti sublattice.

### Fe and Ni in TiAl

Fe and Ni : opposite anisotropy to Ti and In

$$\text{Fe : } D_c / D_a = 1.55-4.09 \quad Q_c \approx Q_a$$

$$\text{Ni: } D_c / D_a = 1.40-3.51 \quad Q_c < Q_a$$

Fe atoms jump on both Ti and Al sublattice.

For Ni diffusion, further investigation is necessary.



## Concluding remark

One can infer diffusion mechanisms in  $L1_0$ -ordered alloys by investigating diffusion anisotropy

For diffusion perpendicular to the [001] axis,

$$D_a = D_a(\text{i}) + D_a(\text{ii, iii}) + D_a(\text{iv})$$

(a) the process (i) or (iv) is predominant

→  $Q_c > Q_a$ ,  $D_c > D_a$  (the case of Ti or Al)

(b) the process (ii, iii) is predominant

→  $Q_c \approx Q_a$ ,  $D_c < D_a$  (the case of Fe)



$Q_c > Q_a$ ,  $D_c > D_a$  → the process (i) or (iv) is predominant.

$Q_c \approx Q_a$ ,  $D_c < D_a$  → the process (ii, iii) is predominant.